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Growth and optical properties of peculiar ZnO tetrapods

Fa-Quan He and Ya-Pu Zhao¹

State Key Laboratory of Non-linear Mechanics (LNM), Institute of Mechanics, Chinese Academy of Sciences, Beijing 100080, People's Republic of China

E-mail: yzhao@lnm.imech.ac.cn

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Abstract

Regular ZnO tetrapods with different morphologies have been obtained on Si(100) substrate via the chemical vapour deposition approach. Varying the growth temperature and gas rate, we have obtained different structured ZnO materials: tetrapods with a large hexagonal crown, a flat top and a small hexagonal crown. The results suggest that these tetrapods are all single crystals with a wurtzite structure that grow along the (0001) direction. However, photoluminescence spectra shows that their optical properties are quite different: for those with large hexagonal crown, the green emission overwhelms that of the near band-edge (NBE) ultraviolet (UV) peak, while others have only a strong NBE UV peak at \sim 386 nm.

1. Introduction

Micro- or nanosize inorganic materials, especially semiconductor materials, have attracted much attention in recent years. As a promising semiconductor with potential applications in light emitting diodes [1,2], transistors [3], sensors [4,5], solar cells [6], optical switches [7], photo-catalysis [8], surface acoustic waveguides [9] and piezoelectric transducers [10], micro/nanosize ZnO has invoked worldwide intensive investigations. Naturally, shape and dimensionality play important roles in the properties and applications of nanomaterials. So far, ZnO nanomaterials with exciting morphologies, such as nanobelts [11], nanowires [12] and nanohelices [13], have been reported. Nowadays, numerous studies have been focused on the preparation and properties of ZnO tetrapods because they possess remarkable optical, electric, magnetic and mechanical properties; thus they have prospective applications in transistors [3], sensors [4, 5], etc.

In this paper, we present a simple method to grow some regular tetrapods at temperatures around 600-700 °C. The as-synthesized products exhibit unique optical properties which shed light on potential applications in the fields of nanoelectronics and nanooptics.

2. Experimental details

These regular ZnO tetrapods were synthesized via a simple catalyst-free vapour evaporation deposition process. P-type

¹ Author to whom any correspondence should be addressed.

at a temperature of 90 °C, then ultrasonically cleaned with deionized water and finally dried naturally. An alumina boat $(8.0 \text{ mm} \times 1.2 \text{ mm} \times 1.0 \text{ mm})$ containing 2.0 g Zn powder (99.999%) was inserted into the centre of the horizontal quartz tube that was placed horizontally in a high-temperature tubular furnace (50 mm in inner diameter and 1200 mm in length). A silicon substrate $(15 \text{ mm} \times 5 \text{ mm} \times 0.38 \text{ mm})$ was laid face-down on the boat with the long edges parallel to the quartz tube. The nearest distance between the edge and the Zn sources was about 6.0 mm. The system was first evacuated for about 10 min, and then Ar was switched on to fill the quartz tube in order to drive off the residual O_2 in it. Subsequently, the sources were heated to the desired temperature (630 °C and 680 °C, respectively) at a rate of 30°C min⁻¹. Immediately, a flow of Ar at a rate of about 300 sccm and O₂ at about 10-30 sccm was introduced into the system and maintained for 10-30 min. Finally, the flows of Ar and O₂ were shut down, and the furnace was turned off and cooled naturally down to room temperature. Products were collected on the surface of the substrates. The as-prepared ZnO tetrapods were characterized using x-ray diffraction (XRD, D/MAX-2500), field emission scanning electron microscopy (FESEM FEI SIRION) equipped with energy-dispersive x-ray spectroscopy (EDS), high-resolution field emission transmission electron microscopy (HR-FETEM, JEM 2010F) equipped with selected area electron diffraction (SAED), Raman spectrum (JY-T64000, excitation wavelength:

silicon (100) wafer was first cleaned with a standard treatment in Piranha solution $(20\% H_2SO_4 + 10\% H_2O_2)$ for about 30 min 524 nm) and the photoluminescence (PL) property (Xe lamp: 325 nm as the excitation source).

3. Results and discussion

To investigate the composition and structure of the obtained samples, the EDS and XRD measurements were employed. The EDS analysis confirms that the synthesized samples are composed of Zn and O with no other elements detected. The XRD results show that the samples synthesized at T = 630 °C and 680 °C are the same as shown in figure 1. As can be seen, all peaks are sharp and strong, suggesting that the products are highly crystallized. These peaks agree well with the peaks of the wurtzite-type ZnO with unit cell constants of a = 3.253 Å and c = 5.209 Å.

Figure 2(*a*) shows typical SEM images of the ZnO tetrapods synthesized at T = 630 °C with a flow of O₂ at a rate of 15 sccm. The as-synthesized products are uniform





tetrapod-like ZnO with a large hexagonal crown on each leg. The legs of each unit, as magnified in figure 2(b) have a diameter of 100–150 nm, a length of about 1–1.5 μ m; the side length of the hexagonal crown is about 500 nm and the thickness around 100 nm. The temperature and flow play an important role in the morphology of the products. When the growth temperature was raised to 680 °C and the flow of O₂ was changed to 25 sccm, two other different structures of ZnO resulted, as shown in figures 2(c)-(f). These two kinds of ZnO tetrapods were obtained from different regions of the same substrate: the ZnO tetrapods with smaller hexagonal crowns shown in figures 2(c) and (d) were collected on the upper side of the Si substrate and the other (shown in figures 2(e)and (f) on the middle side. The four legs of the former tetapods are a remarkably uniform prism with six sides and faces, and each leg is between $1-2 \,\mu m$ in length and about 500 nm in diameter. But the size of each unit may be different from one another. The top shape of the tetrapods is a regular hexagonal crown (smaller than the legs). Figures 2(e) and (f)present a general morphology of the later tetrapods at different magnifications, showing that a unit has four regular legs around $1-2 \mu m$ in length and $\sim 500 nm$ in diameter. Different from the two tetrapods mentioned above, this kind of tetrapod has no crown, and the size of each leg of a unit is smoothly reduced from the top to the centre. A common characteristic for the three kinds of tetrapods is that they have a regular hexagonal top and the cross sections of the legs have an internal angle of 120°.

Further analyses of these ZnO tetrapods were performed using HRFETEM. Typical TEM images of the ZnO tetrapods with large hexagonal crown obtained at a temperature of $630 \,^{\circ}$ C given in figures 3(a) and (b) clearly show that the diameter of a leg is ~120 nm. The HRTEM image as shown in figure 3(c) shows that the spacing between adjacent lattice planes is approximately 0.51 nm corresponding to the (0001) plane of wurtzite-type ZnO. This indicates that the crystals are preferentially oriented in the *c*-axis direction. The SAED patterns confirm that the synthesized products are single



Figure 2. Low and high magnification SEM images of ZnO tetrapods deposited at different temperatures showing the difference of morphology between these regular tetrapods: (*a*), (*b*) $T = 630 \degree \text{C}$, O₂: 15 sccm, Ar: 300 sccm and (*c*)–(*f*) $T = 680 \degree \text{C}$, O₂: 25 sccm, Ar: 300 sccm.



Figure 3. (a), (b), (d), (e) Typical TEM images of three different structured ZnO tetrapods and the upper-right insets of the TEM images correspond to the SEAD patterns indicating the single-crystalline nature of ZnO tetrapods with (0001) orientation; (c) HTEM image of the ZnO tetrapods with large hexagonal crown.



Figure 4. Raman scattering spectrum acquired from the ZnO tetrapods with hexagonal crown at room temperature fabricated at (*a*) T = 630 °C, O₂: 15 sccm, Ar: 300 sccm and (*b*) T = 680 °C, O₂: 25 sccm, Ar: 300 sccm.

crystals which grow along the (0001) orientation. Figures 3(*d*) and (*e*) are the TEM images of the regular ZnO tetrapods fabricated at a temperature of 680 °C, revealing that the ZnO with a flat top is about 2 μ m in length which is longer than those with the small hexagonal crown, but their legs are smaller. Because these two kinds of structured ZnO are too thick to be transparent to the incident electron beam, it is very difficult to obtain the HRTEM images. The SAED patterns taken from one of the legs show that the products are perfect single crystals with wurtzite structure and each leg grows along the (0001) direction, as is consistent with the result of XRD.

Room temperature Raman spectroscopy and PL measurements were performed to examine the optical properties of these regular tetrapods. Typical Raman spectra of these tetrapods are shown in figures 4(*a*) and (*b*). There is not much difference between the two spectra for different structured tetrapods both in intensity and location. Two dominating and strong intensity peaks at ~99.88 cm⁻¹ and ~438.22 cm⁻¹, which are commonly observed in the wurtzite structure ZnO [14], are attributed to the low- and high-E₂ mode of non-polar optical phonons, respectively. Two very small peaks at ~331.6 cm⁻¹ and ~382.53 cm⁻¹ correspond to $E_{2H}-E_{2L}$ (multi phonon) and A_{1T} modes, respectively. The short and suppressed peak at ~578.91 nm is assigned to the E_{1L} mode due to the impurities and structural defects (oxygen vacancies and Zn interstitials) [15] of the samples.

The curve 'a' in figure 5 shows the room temperature PL spectrum of tetrapods with large hexagonal crown. As can be seen, it has two peaks: the near band-edge (NBE) ultraviolet (UV) peak at \sim 393 nm (3.37 eV) and a green emission at \sim 511 nm (2.43 eV). The former peak is considerably weaker compared with the latter, which falls in a very wide range \sim 425– \sim 613 nm. The origin of the green emission is still debatable. Scientists tend to assign it to deep levels induced by the single ionized oxygen vacancy in the ZnO [16]. It could also possibly have resulted from multiple shallow energy levels induced by the defects or their associates because there are some very weak peaks in tetrapods with large hexagonal crowns. The PL spectrum of tetrapods with small hexagonal crowns are shown by curve 'b' in figure 5 (the tetrapods with flat pods have the same spectra). The NBE peak is blue-shifted to 386 nm; besides, it has no pronounced green emission peak except three very weak peaks at about 430, 450 and 470 nm. These indicated that this sample could have defects with very shallow energy level. Maybe, the reaction conditions (e.g. temperature, gas rate) have great influence on the surface state of the synthesized materials. The tetrapods with a large hexagonal crown have high surface-to-volume ratio due to its slender



Figure 5. Room temperature PL spectrum of the ZnO tetrapods excited at 325 nm using Xe lamp as the resource. (a) T = 630 °C, O₂: 15 sccm, Ar: 300 sccm and (b) T = 680 °C, O₂: 25 sccm, Ar: 300 sccm.

legs and the thin hexagonal crown; therefore, there may exist more oxygen vacancies than in other tetrapods. Consequently, the green emission peak at 511 nm is extraordinarily strong and quite wide.

So far, the growth mechanism for the growth of ZnO nanostrutures is still not quite clear. It is generally believed that ZnO tetrapods are formed by the vapour-solid (VS) mechanism rather than by the vapour-liquid-solid (VLS) model. The typical characteristic of the VLS mechanism is that the synthesized materials have metal particles capped at the end of them; however, these peculiar tetrapods were fabricated without the presence of any catalysts or other additives. There are many factors, including growth temperature, time, gas flow, substrate and catalyst, which are of importance to the growth and thus determine the shape and quality of structures. It is reported that the supply rate of oxygen determines the shape and length of tetrapods [17]. In these cases, maybe the temperature, together with the oxygen rate, is mainly responsible for these special structures. In the experiments, it was found that among these peculiar tetrapods there exists a common structured ZnO: peaked-leg tetrapods. Therefore, we can presume that all these regular tetrapods may originate from peaked-leg tetrapods. All experiments were carried out at a low oxygen partial pressure and at temperatures much higher then the melting point of zinc; therefore, the concentration of O₂ is low, while that of Zn is considerably high. For the ZnO tetrapods with a hexagonal crown, the partial pressure of oxygen is very low; the oxygen is consumed quickly in a short period. The sudden change in the reaction atmosphere results in the change in the growth orientation and produces ZnO tetrapods with hexagonal crowns. As for the other two, they are grown at a higher oxygen ratio, so the partial pressure of oxygen does not change so abruptly as to change the growth direction. Of course, the ZnO_x will also gather at the tips of the peak-shaped ZnO for the same reason as mentioned above. They will grow

layer by layer according to the symmetry of the ZnO crystal and finally result in very regular legs. But, in different regions of the boat, there is little difference in the reaction conditions, such as temperature and gas rate, determining the difference in configuration of the products in different regions. However, the growth mechanism mentioned is just a speculated result and it needs further support from experiment and theory.

4. Conclusions

In summary, three different kinds of peculiar structured ZnO tetrapods have been successfully synthesized by changing the growth temperature. SAED pattern and XRD results suggest that they are single crystals with wurtzite structure and grow along the (0001) direction. PL spectra of these structured ZnO reveal that they have unique optical properties, which suggest potential applications in the field of nanodevices.

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References

- Ryu Y R, Zhu S, Budai J D, Chanandrasekhar H R, Miceli P F and White H W 2000 J. Appl. Phys. 88 201
- [2] Service R F 1997 Science 276 895
- [3] Arnold M S, Avouris P, Pan Z W and Wang Z L 2003 J. Phys. Chem. B 107 659
- [4] Cui Y, Wei Q Q, Park H K and Lieber C M 2001 Science 293 17
- [5] Sekhar R C, Hari K S, Vivekchand S R C, Govindaraj A and Rao C N R 2005 Chem. Phys. Lett. 418 582
- [6] O'Regan B, Schwarthz D T, Zakeeruddin S M and Gratzel M 2000 Adv. Mater. 12 1263
- [7] Nagata T, Shimura T, Asida A, Fujimura N and Ito T 2002 J. Cryst. Growth 273 533
- [8] Yumoto H, Inoue Y, Lee S J, Sako T and Nishiyama K 1999 Thin Solid Films 345 38
- [9] Chen J J, Zeng F, Li D M, Niu J B and Pan F 2005 Thin Solid Films 485 257
- [10] Fortunato E, Barquinha P, Pimentel A, Gonçalves A, Marques A, Pereira L and Martins R 2005 *Thin Solid Films* 487 205
- [11] Pan Z W, Dai Z R and Wang Z L 2001 Science 291 1947
- [12] Kong Y C, Yu D P, Zhang B, Fang W and Feng S Q 2001 Appl. Phys. Lett. 78 407
- [13] Gao P X, Ding Y, Mai W J, Hughes W L, Lao C S and Wang Z L 2005 Science 309 1702
- [14] Xing Y J, Xi Z H, Xue Z Q, Zhang X D, Song J H, Wang R M, Xu J, Song Y, Zhang S L and Yu D P 2003 Appl. Phys. Lett. 83 1689
- [15] Vanheusden K, Seager C H, Warren W L, Tallant D R and Voigt J A 1996 J. Appl. Phys. 79 7983
- [16] Heo Y W, Norton D P and Pearton S J 2005 J. Appl. Phys. 98 073502
- [17] Kitano M, Hamabe T and Maeda S 1990 J. Cryst. Growth 102 965